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SOFT CONTACT LENSES

RELATED U.S. APPLICATIONS

This application is a continuation-in-part of U.S. Serial No. 09/652,817,
5 filed on August 30, 2000, which is a continuation-in-part of U.S. Serial No.
09/532,943, filed on March 22, 2000, which is a continuation-in-part of U.S.
Serial No. 09/414,365, filed on October 7, 1999, which is a continuation-in-part
of U.S. Serial No. 09/033,347, filed on March 2, 1998, now issued as U.S.
Patent 5,998,498.

10 FIELD OF THE INVENTION

This invention relates to silicone hydrogels. In particular, the invention
relates to silicone hydrogels formed by curing a reaction mixture of silicone-
containing monomers.

15 BACKGROUND OF THE INVENTION

A hydrogel is a hydrated crosslinked polymeric system that contains
water in an equilibrium state. Hydrogels typically are oxygen permeable and
biocompatible, making them preferred materials for producing biomedical
devices and in particular contact or intraocular lenses.

20 Conventional hydrogels are prepared from monomeric mixtures
predominantly containing hydrophilic monomers, such as 2-hydroxyethyl
methacrylate ("HEMA") or N-vinyl pyrrolidone ("NVP"). United States Patents
Nos. 4,495,313, 4,889,664 and 5,039,459 disclose the formation of
conventional hydrogels. The oxygen permeability of these conventional
25 hydrogel materials relates to the water content of the materials, and is typically
below 20-30 barrers. For contact lenses made of the conventional hydrogel
materials, that level of oxygen permeability is suitable for short-term wear of
the contact lenses; however, that level of oxygen permeability may be
insufficient to maintain a healthy cornea during long-term wear of contact
30 lenses (e.g., 30 days without removal). Therefore, efforts have been made and
continue to be made to increase the oxygen permeability of conventional
hydrogels.

One known way to increase the oxygen permeability of hydrogels is to add silicone-containing monomers to the hydrogel formulations to produce silicone hydrogels. Silicone-containing polymers generally have higher oxygen permeabilities than conventional hydrogels. Silicone hydrogels have typically been prepared by polymerizing mixtures containing at least one silicone-containing monomer and at least one hydrophilic monomer. Either the silicone-containing monomer or the hydrophilic monomer may function as a crosslinking agent (a crosslinking agent is a monomer having multiple polymerizable functionalities) or a separate crosslinking agent may be employed. The formation of silicone hydrogels has been disclosed in U.S. Patents Nos. 4,954,587, 5,010,141, 5,079,319, 5,115,056, 5,260,000, 5,336,797, 5,358,995, 5,387,632, 5,451,617, 5,486,579, WO 96/31792, US 5789461, US 5776999, US 5760100 and US 5849811. Group Transfer Polymerization techniques for polymerizing acrylic and methacrylic monomers with terminal silyl containing monomers is described in various patents including U.S. Patents Nos. 4,414,372, 4,417,034, 4,508,880, 4,524,196, 4,581,428, 4,588,795, 4,598,161, 4,605,716, 4,622,372, 4,656,233, 4,659,782, 4,659,783, 4,681,918, 4,695,607, 4,711,942, 4,771,116, 5,019,634 and 5,021,524 each of which is incorporated in its entirety herein by reference.

U.S. Patent No. 3,808,178 discloses the formation of copolymers of small silicone-containing monomers and various hydrophilic monomers. U.S. Patent No. 5,034,461 describes silicone hydrogels prepared from various combinations of silicone-polyurethane macromers and hydrophilic monomers such as HEMA or N,N-dimethacrylamide ("DMA"). The addition of methacryloxypropyltris-(trimethylsiloxy)silane ("TRIS") reduced the modulus of such hydrogels, but in many examples the modulus was still higher than may be desired.

U.S. Patents Nos. 5,358,995 and 5,387,632 describe hydrogels made from various combinations of silicone macromers, TRIS, NVP and DMA.

Replacing a substantial portion of the silicone macromer with TRIS reduced the modulus of the resulting hydrogels. Two publications from the same author,

"The Role of Bulky Polysiloxanylalkyl Methacrylates in Polyurethane-Polysiloxane Hydrogels", J. Appl. Poly. Sci., Vol. 60, 1193-1199 (1996), and "The Role of Bulky Polysiloxanylalkyl Methacrylates in Oxygen-Permeable Hydrogel Materials", J. Appl. Poly. Sci., Vol. 56, 317-324 (1995) also describe experimental results indicating that the modulus of hydrogels made from reaction mixtures of silicone-macromers and hydrophilic monomers such as DMA decreases with added TRIS.

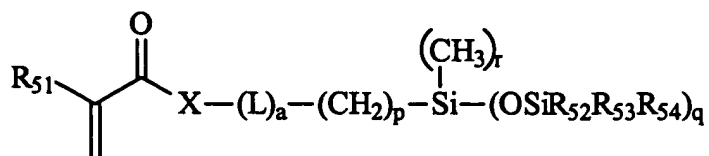
The use of methacryloxypropylbis(trimethylsiloxy)methylsilane ("MBM") to make hard contact lenses was described in WO 9110155 and in JP 61123609.

When relatively high levels of bulky silicone-containing monomers such as TRIS are incorporated into the hydrogels made from silicone-containing macromers and hydrophilic monomers, time at which the polymer returns to its original shape after applied stress is relieved increases to an extent that is unacceptable to the contact lens wearer.

There still remains a need in the art for silicone hydrogels that are soft enough to make soft contact lenses, which possess high oxygen permeability, suitable water content, and sufficient elasticity, and are comfortable to the contact lens wearer.

SUMMARY OF THE INVENTION

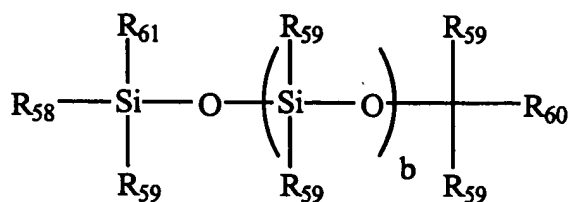
This invention provides a silicone hydrogel prepared by curing a reaction mixture comprising either or both of the silicone-containing monomers of Structure I and II. Structure I has the following structure:



wherein R_{51} is a monovalent group such as H, C_{1-5} alkyl, or an ethylenically unsaturated moiety (such as styryl, C_{1-5} alkenyl and the like) where H, CH_3 are preferred, q is 1, 2, or 3 and for each q , R_{52} , R_{53} and R_{54} are independently alkyl

or aromatic, preferably ethyl, methyl, benzyl, phenyl, or a monovalent siloxane chain comprising from 1 to 100 repeating Si-O units, p is 1 to 10, r = (3-q), X is O or NR₅₅, where R₅₅ is H or a monovalent alkyl group with 1 to 4 carbons, a is 0 or 1, and L is a divalent linking group which preferably comprises from 2 to 5
 5 carbons, which may also optionally comprise ether or hydroxyl groups, for example, a polyethylene glycol chain.

Structure II has the following structure:



10

where b = 0 to 100, where it is understood that b is a distribution having a mode equal to a stated value, preferably 8 to 10; R₅₈ is a monovalent group containing at least one ethylenically unsaturated moiety, preferably a
 15 monovalent group containing a styryl, vinyl, or methacrylate moiety, more preferably a methacrylate moiety; each R₅₉ is independently a monovalent alkyl, or aryl group, which may be further substituted with alcohol, amine, ketone, carboxylic acid or ether groups, preferably unsubstituted monovalent alkyl or aryl groups, more preferably methyl; R₆₀ is a monovalent alkyl, or aryl
 20 group, which may be further substituted with alcohol, amine, ketone, carboxylic acid or ether groups, preferably unsubstituted monovalent alkyl or aryl groups, preferably a C₁₋₁₀ aliphatic or aromatic group which may include hetero atoms, more preferably C₃₋₈ alkyl groups, most preferably butyl, and R₆₁ is
 25 monovalent siloxane chain comprising from 1 to 100 repeating Si-O units.

In the preferred embodiment, the silicone hydrogel comprises monomers of both Structure I and II. More preferably, the silicone hydrogel

comprises silicone-containing monomer of Structure I and II and a hydrophilic monomer.

Among the advantages of this invention is that the use of the silicone-containing monomers of both Structure I and Structure II in a silicone hydrogel
5 reduces the Young's modulus of the hydrogel especially in hydrogels which comprise these silicone-containing monomers and additional silicone-containing monomers which act as crosslinkers. The monomers of Structure I and II are more effective at lowering the modulus of the silicone hydrogel than for monomers described in the prior art. Additionally, the $\tan \delta$ of the silicone
10 hydrogels of this invention may be concurrently preserved.

The polymers produced according to this invention can be used to produce soft contact lenses that will provide high oxygen permeability, good elasticity, and can be produced economically and efficiently. The polymer of this invention can be used to make biomedical devices which require
15 biocompatibility and high oxygen permeability, preferably contact lenses.

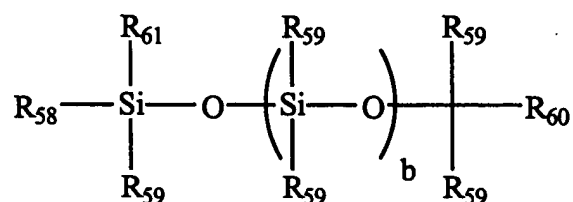
DETAILED DESCRIPTION OF THE INVENTION AND PREFERRED EMBODIMENTS

The term "monomer" used herein refers to lower molecular weight compounds
20 that can be polymerized to higher molecular weight compounds, polymers, macromers, or prepolymers. The term "macromer" as used herein refers to a high molecular weight polymerizable compound. Prepolymers are partially polymerized monomers or monomers which are capable of further polymerization.

25 A "silicone-containing monomer" is one that contains at least two [-Si-O-] repeating units in a monomer, macromer or prepolymer. Preferably, the total Si and attached O are present in the silicone-containing monomer in an amount greater than 20 weight percent, and more preferably greater than 30 weight percent of the total molecular weight of the silicone-containing
30 monomer.

Examples of the silicone-containing monomers of Structure I that can be used to form silicone hydrogels of this invention are, without limitation, methacryloxypropylbis(trimethylsiloxy)methylsilane, methacryloxypropyltris(trimethylsiloxy)silane,
 5 methacryloxypropylpentamethyldisiloxane, and (3-methacryloxy-2-hydroxypropyloxy) propylbis(trimethylsiloxy)methylsilane. While such silicone monomers may additionally be used, linear mono-alkyl terminated polydimethylsiloxanes ("mPDMS") such as those shown in the following Structure II must be used:

10



where $b = 0$ to 100 , where it is understood that b is a distribution having a mode equal to a stated value, preferably 4 to 16 , more preferably 8 to 10 ; R_{58} is
 15 a monovalent group containing at least one ethylenically unsaturated moiety, preferably a monovalent group containing a styryl, vinyl, or methacrylate moiety, more preferably a methacrylate moiety; each R_{59} is independently a monovalent alkyl, or aryl group, which may be further substituted with alcohol, amine, ketone, carboxylic acid or ether groups, preferably unsubstituted
 20 monovalent alkyl or aryl groups, more preferably methyl; R_{60} is a monovalent alkyl, or aryl group, which may be further substituted with alcohol, amine, ketone, carboxylic acid or ether groups, preferably unsubstituted monovalent alkyl or aryl groups, preferably a C_{1-10} aliphatic or aromatic group which may include hetero atoms, more preferably C_{3-8} alkyl groups, most preferably butyl;
 25 and R_{61} is independently alkyl or aromatic, preferably ethyl, methyl, benzyl, phenyl, or a monovalent siloxane chain comprising from 1 to 100 repeating Si-O units.

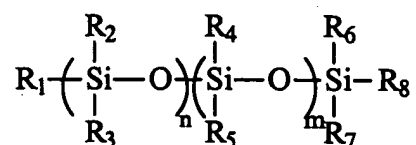
The amount of mPDMS comprising the hydrogel is closely related to the modulus and $\tan \delta$ of the hydrogels made according to this invention. $\tan \delta$ is defined as the loss modulus of the material divided by its elastic modulus (G''/G'). For purposes of this invention, modulus is defined as Young's

5 modulus or its equivalent tensile modulus where the measurement is recorded at the equilibrated hydrated state. It is desirable to lower both the modulus and $\tan \delta$ in silicone hydrogel lenses for a number of reasons. First, lower modulus and $\tan \delta$ in a lens are manifested as less stiffness, and after stress is relieved the lens quickly returns to its original shape. This improves comfort over
10 traditional silicone hydrogel lenses and makes them more aesthetically appealing given their ability to retain their round shape. Further, the incidence of superior epithelial arcuate lesions ("SEALs") is either or both lessened and eliminated by using lenses made from a polymer having a sufficiently low modulus and $\tan \delta$. Thus, replacing lenses made from high modulus, high \tan
15 d polymers with those of the instant invention is a means for reducing or eliminating the occurrence of SEALs. This is particularly the case for contact lens wearers that are prone to SEALs.

Desirably, silicone hydrogels made according to the invention comprise between about 2 and 70%wt mPDMS based on total weight of reactive
20 monomer components from which the polymer is made. Depending upon the other monomers present, this will generally reduce the modulus of the polymer to between about 20 and 180 psi and a $\tan \delta$ of less than about 0.1 to no more than about 0.3 (measured at a frequency of 1 Hz and a temperature of 25°C, according to the method described in Example 21). Silicone hydrogels made
25 according to the invention and comprising between about 4 and 50 %wt mPDMS (same basis as above) are preferred. These will generally exhibit a modulus between about 30 and 160 psi and a $\tan \delta$ of about 0.05 to about 0.3 (measured at a frequency of 1 Hz and a temperature of 25°C). Silicone hydrogels made according to the invention and comprising between about 8
30 and 40 % wt mPDMS (same basis as above) are most preferred. These hydrogels will generally exhibit a modulus between about 40 and 130 psi and a

tan δ of about 0.2 or less (measured at a frequency of 1 Hz and a temperature of 25°C). Hydrogels having tan δ less than about 0.1 can also be made according to this invention as described more fully below.

Additional silicone-containing monomers may be combined with the
 5 silicone-containing monomers of Structures I and II to form the soft contact lenses of the invention. Any known silicone-containing monomers useful for making silicone hydrogels can be used in combination with the silicone-containing monomer of Structure I and II to form the soft contact lenses of this invention. Many silicone-containing monomers useful for this purpose are
 10 disclosed in U.S. Patent No. 6,020,445 incorporated herein in its entirety by reference. Useful additional silicone-containing monomers combined with the silicone-containing monomers of Structure I to form the silicone hydrogels of this invention are the hydroxyalkylamine-functional silicone-containing monomers disclosed in U.S. Patent No. 5,962,548 incorporated herein in its
 15 entirety by reference. The preferred silicone-containing linear or branched hydroxyalkylamine-functional monomers comprising a block or random monomer of the following structure:



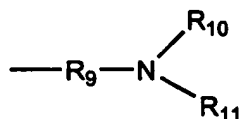
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Structure III

wherein:

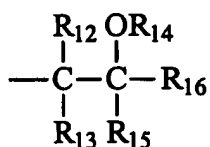
n is 0 to 500 and m is 0 to 500 and (n + m) = 10 to 500 and more preferably 20
 25 to 250; R_2 , R_4 , R_5 , R_6 and R_7 are independently a monovalent alkyl, or aryl groups, which may be further substituted with alcohol, ester, amine, ketone, carboxylic acid or ether groups, preferably unsubstituted monovalent alkyl or aryl groups; and R_1 , R_3 and R_8 are independently a monovalent alkyl, or aryl group, which may be further substituted with an alcohol, ester, amine, ketone,

carboxylic acid or ether group, preferably unsubstituted monovalent alkyl or aryl groups, or are the following nitrogen-containing structure:



Structure IV

- 5 with the proviso that at least one of R_1 , R_3 , and R_8 are according to Structure IV, wherein R_9 is a divalent alkyl group such as $-(CH_2)_s-$ where s is from 1 to 10, preferably 3 to 6 and most preferably 3;
 R_{10} and R_{11} are independently H, a monovalent alkyl or aryl group which may be further substituted with an alcohol, ester, amine, ketone, carboxylic acid or
 10 ether group, or has the following structure:



Structure V

- where R_{14} is H, or a monovalent polymerizable group comprising acryloyl, methacryloyl, styryl, vinyl, allyl or N-vinyl lactam, preferably H or
 15 methacryloyl; R_{16} is either H, a monovalent alkyl or aryl group which can be further substituted with alcohol, ester, amine, ketone, carboxylic acid or ether groups, or a polymerizable group comprising acrylate, methacrylate, styryl, vinyl, allyl or N-vinyl lactam, preferably alkyl substituted with an alcohol or methacrylate; R_{12} , R_{13} and R_{15} are independently H, a monovalent alkyl or aryl,
 20 which can be further substituted with alcohol, ester, amine, ketone, carboxylic acid or ether groups, or R_{12} and R_{15} , or R_{15} and R_{13} can be bonded together to form a ring structure, with the proviso that at least some of the Structure IV groups on the monomer comprises polymerizable groups. R_{12} , R_{13} and R_{15} are preferably H.

- 25 In alternative embodiments, the silicone hydrogels of this invention, comprising the silicone-containing monomers of either or both Structure I and Structure II may further comprise hydrophilic monomers. The hydrophilic monomers optionally used to make the hydrogel polymer of this invention can

be any of the known hydrophilic monomers disclosed in the prior art to make hydrogels.

The preferred hydrophilic monomers used to make the polymer of this invention may be either acrylic- or vinyl-containing. Such hydrophilic monomers may themselves be used as crosslinking agents. The term "vinyl-type" or "vinyl-containing" monomers refer to monomers containing the vinyl grouping ($-\text{CH}=\text{CH}_2$) and are generally highly reactive. Such hydrophilic vinyl-containing monomers are known to polymerize relatively easily.

"Acrylic-type" or "acrylic-containing" monomers are those monomers containing the acrylic group: ($\text{CH}_2=\text{C}(\text{R})\text{COX}$) wherein R is H or CH_3 , and X is O or N, which are also known to polymerize readily, such as N,N-dimethyl acrylamide (DMA), 2-hydroxyethyl methacrylate (HEMA), glycerol methacrylate, 2-hydroxyethyl methacrylamide, polyethyleneglycol monomethacrylate, methacrylic acid and acrylic acid.

Hydrophilic vinyl-containing monomers which may be incorporated into the silicone hydrogels of the present invention include monomers such as N-vinyl lactams (e.g. NVP), N-vinyl-N-methyl acetamide, N-vinyl-N-ethyl acetamide, N-vinyl-N-ethyl formamide, N-vinyl formamide, with NVP being preferred.

Other hydrophilic monomers that can be employed in the invention include polyoxyethylene polyols having one or more of the terminal hydroxyl groups replaced with a functional group containing a polymerizable double bond. Examples include polyethylene glycol, ethoxylated alkyl glucoside, and ethoxylated bisphenol A reacted with one or more molar equivalents of an end-capping group such as isocyanatoethyl methacrylate ("IEM"), methacrylic anhydride, methacryloyl chloride, vinylbenzoyl chloride, or the like, to produce a polyethylene polyol having one or more terminal polymerizable olefinic groups bonded to the polyethylene polyol through linking moieties such as carbamate or ester groups.

Still further examples are the hydrophilic vinyl carbonate or vinyl carbamate monomers disclosed in U.S. Patents No. 5,070,215, and the

hydrophilic oxazolone monomers disclosed in U.S. Patents No. 4,910,277.
Other suitable hydrophilic monomers will be apparent to one skilled in the art.

More preferred hydrophilic monomers which may be incorporated into the polymer of the present invention include hydrophilic monomers such as
5 DMA, HEMA, glycerol methacrylate, 2-hydroxyethyl methacrylamide, NVP, polyethyleneglycol monomethacrylate, methacrylic acid and acrylic acid with DMA being the most preferred.

Other monomers that can be present in the reaction mixture used to form the silicone hydrogel of this invention include ultra-violet absorbing
10 monomers, reactive tints, pigments, and the like. Additional processing aids such as release agents or wetting agents can also be added to the reaction mixture.

A polymerization initiator is preferably included in the reaction mixture. The polymerization initiator can be a compound such as lauryl peroxide,
15 benzoyl peroxide, isopropyl percarbonate, azobisisobutyronitrile, or the like, that generates free radicals at moderately elevated temperatures, or the polymerization initiator can be a photoinitiator system such as an aromatic alpha-hydroxy ketone or a tertiary amine plus a diketone. Illustrative examples of photoinitiator systems are 2-hydroxy-2-methyl-1-phenyl-propan-1-one, and a
20 combination of camphorquinone and ethyl 4-(N,N-dimethylamino)benzoate. The initiator is used in the reaction mixture in effective amounts, e.g., from about 0.1 to about 2 parts by weight per 100 parts of reactive monomer. Polymerization of the reaction mixture can be initiated using the appropriate choice of heat or visible or ultraviolet light or other means depending on the
25 polymerization initiator used. The preferred initiator is a 1:1 blend of 1-hydroxycyclohexyl phenyl ketone and bis(2, 6-dimethoxybenzoyl)-2, 4, 4-trimethylpentyl phosphine oxide and the preferred method of polymerization initiation is visible light.

Typically after curing of the reaction mixture of the silicone-containing
30 monomers of either or both Structure I and II and optional hydrophilic monomers and any other optional ingredients such as additional silicone-

containing monomers, diluents, crosslinking agents, catalysts, release agents, tints etc. which are blended together prior to polymerization, the resulting polymer is treated with a solvent to remove the diluent (if used) or any traces of unreacted components, and hydrate the polymer to form the hydrogel. The solvent used may be water (or an aqueous solution such as physiological saline), or depending on the solubility characteristics of the diluent (if used) used to make the hydrogel of this invention and the solubility characteristics of any residual unpolymerized monomers, the solvent initially used can be an organic liquid such as ethanol, methanol, isopropanol, mixtures thereof, or the like, or a mixture of one or more such organic liquids with water, followed by extraction with pure water (or physiological saline) to produce the silicone hydrogel comprising a polymer of said monomers swollen with water. The silicone hydrogels after hydration of the polymers preferably comprise about 2 to 50 weight percent water, more preferably about 15 to 45 weight percent water, and most preferably about 20 to 40 weight percent water of the total weight of the silicone hydrogel. These silicone hydrogels are particularly suited for making contact lenses or intraocular lenses, preferably soft contact lenses.

In another preferred embodiment, a silicone hydrogel lens is made by reacting a macromer with a reaction mixture that includes silicone based monomers and hydrophilic monomers. This technique affords a high level of control of the structure of the ultimate product. Phase distribution can be controlled so that a more uniform coating or surface layer (if desired) can be applied to the lens. By "surface layer" is meant a distribution of material with a portion in contact with the environment and another portion in contact with a material having a different bulk property than that of the material from which the surface layer is formed. Additionally, it is easier to process the lenses because of greater uniformity of properties across the lens.

The macromers are made by combining a/an (meth)acrylate and a silicone in the presence of a Group Transfer Polymerization ("GTP") catalyst. These macromers typically comprise copolymers of various monomers. They may be formed in such a way that the monomers come together in distinct

blocks, or in a generally random distribution. These macromers may furthermore be linear, branched, or star shaped. Branched structures are formed for instance if polymethacrylates, or crosslinkable monomers such as ethyleneglycol dimethacrylate are included in the macromer. Initiators, reaction
5 conditions, monomers, and catalysts that can be used to make GTP polymers are described in "Group-Transfer Polymerization" by O.W. Webster, in Encyclopedia of Polymer Science and Engineering Ed. (John Wiley & Sons) p. 580, 1987. These polymerizations are conducted under anhydrous conditions. Hydroxyl-functional monomers, like HEMA, can be incorporated as their
10 trimethylsiloxy esters, with hydrolysis to form free hydroxyl group after polymerization. GTP offers the ability to assemble macromers with control over molecular weight distribution and monomer distribution on the chains. This macromer is then reacted with a reaction mixture comprising predominantly polydimethylsiloxane (preferably, mPDMS), and hydrophilic monomers.

15 Preferred macromer components include mPDMS, TRIS, methyl methacrylate, HEMA, DMA, methacrylonitrile, ethyl methacrylate, butyl methacrylate, 2-hydroxypropyl-1-methacrylate, 2-hydroxyethyl methacrylamide and methacrylic acid. It is even more preferred that the macromer is made from a reaction mixture comprising HEMA, methyl methacrylate, TRIS, and
20 mPDMS. It is most preferred that macromer is made from a reaction mixture comprising, consisting essentially of, or consisting of about 19.1 moles of blocked HEMA (2-(trimethylsiloxy)ethyl methacrylate) about 2.8 moles of methyl methacrylate, about 7.9 moles of TRIS, and about 3.3 moles of mono-methacryloxypropyl terminated mono-butyl terminated polydimethylsiloxane,
25 and is completed by reacting the aforementioned material with about 2.0 moles per mole of 3-isopropenyl- ω,ω -dimethylbenzyl isocyanate using dibutyltin dilaurate as a catalyst.

Silicone hydrogels can be made by reacting blends of macromers, monomers, and other additives such as UV blockers, tints, polymerization
30 inhibitors, and internal wetting agents. Internal wetting agents are substances that are incorporated into the polymer blend of a lens prior to polymerization

and due to their incorporation, the wettability of the lens increases. The reactive components of these blends typically comprise a combination of hydrophobic silicone with hydrophilic components. Since these components are often immiscible because of their differences in polarity, it is particularly
5 advantageous to incorporate a combination of hydrophobic silicone monomers with hydrophilic monomers, especially those with hydroxyl groups, into the macromer. The macromer can then serve to compatibilize the additional silicone and hydrophilic monomers that are incorporated in the final reaction mixture. These blends typically also contain diluents to further compatibilize
10 and solubilize all components. Preferably, the silicone based hydrogels are made by reacting the following monomer mix: macromer; an Si_{8-10} monomethacryloxy terminated polydimethyl siloxane; and hydrophilic monomers together with minor amounts of additives and photoinitiators. It is more preferred that the hydrogels are made by reacting macromer; an Si_{8-10}
15 monomethacryloxy terminated polydimethyl siloxane; TRIS; DMA; HEMA; and tetraethyleneglycol dimethacrylate ("TEGDMA"). It is most preferred that the hydrogels are made from the reaction of (all amounts are calculated as weight percent of the total weight of the combination) macromer (about 18%); an Si_{8-10} monomethacryloxy terminated polydimethyl siloxane (about 28%); TRIS (about
20 14%); DMA (about 26%); HEMA (about 5%); TEGDMA (about 1%), polyvinylpyrrolidone ("PVP") (about 5%); with the balance comprising minor amounts of additives and photoinitiators, and that the reaction is conducted in the presence of 20%wt 3,7-dimethyl-3-octanol diluent.

Various processes are known for molding the reaction mixture in the
25 production of contact lenses, including spincasting and static casting. Spincasting methods are disclosed in U.S. Patents Nos. 3,408,429 and 3,660,545, and static casting methods are disclosed in U.S. Patents Nos. 4,113,224 and 4,197,266. The preferred method for producing contact lenses comprising the polymer of this invention is by the direct molding of the silicone
30 hydrogels, which is economical, and enables precise control over the final shape of the hydrated lens. For this method, the reaction mixture is placed in a

mold having the shape of the final desired silicone hydrogel, i.e. water-swollen polymer, and the reaction mixture is subjected to conditions whereby the monomers polymerize, to thereby produce a polymer in the approximate shape of the final desired product. Then, this polymer mixture is

5 optionally treated with a solvent and then water, producing a silicone hydrogel having a final size and shape which are quite similar to the size and shape of the original molded polymer article. This method can be used to form contact lenses and is further described in U.S. Patents Nos. 4,495,313; 4,680,336; 4,889,664; and 5,039,459, incorporated herein by reference. After producing
10 the silicone hydrogel, the lens may be coated with a hydrophilic coating. Some methods of adding hydrophilic coatings to a lens have been disclosed in the prior art, including U.S. Patents Nos. 3,854,982, 3,916,033, 4,920,184, 5,002,794, 5,779,943, 6,087,415; WO 91/04283, and EPO 93/810,399.

The preferred range of the combined silicone-containing monomer of
15 Structure II and additional silicone-containing monomers, if present in the reaction mixture, is from about 5 to 100 weight percent, more preferably about 10 to 90 weight percent, and most preferably about 15 to 80 weight percent of the reactive components in the reaction mixture. The preferred range of optional hydrophilic monomer if present in the above invention is from about 5
20 to 80 weight percent, more preferably about 10 to 60 weight percent, and most preferably about 20 to 50 weight percent of the reactive components in the reaction mixture. The preferred range of diluent is from about 0 to 70 weight percent, more preferably about 0 to 50 weight percent, and most preferably about 0 to 20 weight percent of the total reaction mixture. The amount of
25 diluent required varies depending on the nature and relative amounts of the reactive components.

In a preferred combination of reactive components about 10 to 60, more preferably about 15 to 50 weight percent of the reactive components is silicone-containing monomer, about 20 to 50 weight percent of the reactive
30 components is silicone-containing monomer of Structure I or Structure II, about 10 to 50 percent of the reactive components is a hydrophilic monomer, more

preferably DMA, about 0.1 to 1.0 percent of the reactive components is a UV or visible light-active photoinitiator and about 0 to 20 weight percent of the total reaction mixture is a secondary or tertiary alcohol diluent, more preferably a tertiary alcohol.

5 The reaction mixtures of the present invention can be formed by any of the methods known to those skilled in the art, such as shaking or stirring, and used to form polymeric articles or devices by the methods described earlier. For some monomer reaction mixtures it is preferred to polymerize the reaction mixtures at temperatures such as 30-100°C, more preferably 50 to 80 °C or 60
10 -70°C.

 Silicone hydrogels of the instant invention have high oxygen permeability. They have O₂ Dk values between about 40 and 300 barrer determined by the polarographic method. Polarographic method measurements of oxygen permeability are made as follows. Lenses are
15 positioned on the sensor then covered on the upper side with a mesh support. The oxygen which diffuses through the lens is measured using a polarographic oxygen sensor consisting of a 4 mm diameter gold cathode and a silver ring anode. The reference values are those measured on commercially available contact lenses using this method. Balafilcon A lenses available from Bausch &
20 Lomb give a measurement of approximately 79 barrer (1 barrer = 10⁻¹⁰ (cm³ of gas x cm²)/(cm³ of polymer x s x cm Hg). Etafilcon A lenses give a measurement of about 20 to 25 barrer.

 Contact lenses made from the silicone hydrogels of the invention may be produced to include a hydrophilic surface layer. Suitable materials for
25 forming the surface layer are known in the art. Preferred materials include poly(vinyl alcohol), polyethylene oxide, poly(2-hydroxyethyl methacrylate), poly(methyl methacrylate), poly(acrylic acid), poly(methacrylic acid), poly(maleic acid), poly(itaconic acid), poly(acrylamide), poly(methacrylamide), poly(dimethylacrylamide), poly(glycerol methacrylate), polystyrene sulfonic
30 acid, polysulfonate polymers, poly(vinyl pyrrolidone), carboxymethylated polymers, such as carboxymethylcellulose, polysaccharides, glucose amino

glycans, polylactic acid, polyglycolic acid, block or random copolymers of the
aforementioned, and the like, and mixtures thereof. Preferably, the carboxyl
functional hydrophilic polymer is poly(acrylic acid), poly(methacrylic acid),
poly(meth)acrylamide, or poly(acrylamide). More preferably, poly(acrylic acid)
5 or poly(acrylamide) is used. Methods for coating contact lenses are disclosed
in U.S. patent No. 6,087,415, and WO 00127662 incorporated herein in their
entirety by reference.

The non-limiting examples below further describe this invention. In the
examples the following abbreviations are used:

10

Examples

	MBM	3-methacryloxypropylbis(trimethylsiloxy)methylsilane
	MPD	methacryloxypropylpentamethyl disiloxane
	TRIS	3-methacryloxypropyltris (trimethylsiloxy) silane
15	DMA	N,N-dimethylacrylamide
	THF	tetrahydrofuran
	TMI	dimethyl meta-isopropenyl benzyl isocyanate
	HEMA	2-hydroxyethyl methacrylate
	TEGDMA	tetraethyleneglycol dimethacrylate
20	EGDMA	ethyleneglycol dimethacrylate
	MMA	methyl methacrylate
	TBACB	tetrabutyl ammonium-m-chlorobenzoate
	mPDMS	monomethacryloxypropyl terminated polydimethylsiloxane (MW 800-1000 unless otherwise indicated)
25	PDMS	polydimethylsiloxane
	3M3P	3-methyl-3-pentanol
	Norbloc	2-(2'-hydroxy-5-methacryloxyethylphenyl)-2H-benzotriazole
	CGI 1850	1:1 (wt) blend of 1-hydroxycyclohexyl phenyl ketone and bis(2,6- dimethoxybenzoyl)-2,4,4-trimethylpentyl phosphine oxide
30	PAA	poly (acrylic acid)
	PVP	poly(N-vinyl pyrrolidone)

IPA isopropyl alcohol
DAROCUR 1173 2-hydroxy-2-methyl-1-phenyl-propan-1-one
D3O 3, 7-dimethyl-3-octanol
HOAc acetic acid
5 TAA t-amyl alcohol
blocked Hema 2-(trimethylsiloxy)ethyl methacrylate
Blue HEMA the reaction product of reactive blue number 4 and HEMA,
as described in Example 4 or U.S. Pat. no. 5,944,853

10 PREPARATION 1 – Preparation of Polysiloxane Macromer

500 grams of α,ω -bisaminopropyl polydimethylsiloxane (5000 MW) and
68 grams of glycidyl methacrylate were combined and heated with stirring at
100°C for 10 hours. The product was extracted five times with 1500 ml of
acetonitrile to remove residual glycidyl methacrylate to give a clear oil. IR:
15 3441, 2962, 1944, 1725, 1638, 1612, 1412 cm^{-1} . This product will be referred
to as "Prep 1" or alternatively bis(N,N-bis-2-hydroxy-3-
methacryloxypropyl)aminopropyl polydimethylsiloxane.

Example 1

20 38.2 parts by weight of the product of PREPARATION 1 was combined
with 28.8 parts MBM, 33 parts DMA and 1 part DAROCUR 1173 and diluted
with 3-methyl-3-pentanol to make a reaction mixture in which the diluent made
up 9% of the mass of the complete reaction mixture. The resulting reaction
mixture was a clear, homogeneous solution. Polypropylene contact lens molds
25 were filled, closed and irradiated with a total of 3.2 J/cm^2 UV light from a
fluorescent UV source over a 30-minute period. The molds were opened and
the lenses were released into isopropanol and then transferred into deionized
water.

The lenses were clear and had a tensile modulus of 205 ± 12 psi, an
30 elongation at break of 133 ± 37 %, and an equilibrium water content of 24.2
 ± 0.2 %. Tensile properties were determined using an Instron™ model 1122

tensile tester (tensile modulus is equivalent to Young's modulus). Equilibrium Water Contents (EWC) were determined gravimetrically and are expressed as:
 $\%EWC = 100 \times (\text{mass of hydrated lens} - \text{mass of dry lens}) / \text{mass of hydrated lens}$

5

Examples 2-16

Reaction mixtures were made using the formulation of Example 1, but with amounts listed in Table 1. All reaction mixtures and lenses were clear.

Table 1 – Silicone Hydrogel Formulations and Properties.

Example	1	2	3	4
Composition				
Prep 1	38.2	33.5	27.6	22.3
MBM	28.8	33.5	39.4	44.7
DMA	33	33	33	33
Darocur 1173	0.4	0.4	0.4	0.4
% of Diluent *	9	7	5	4
EWC(%)	24.2 ± 0.2	23.3 ± 0.3	22.4 ± 0.2	24.2 ± 0.3
Modulus (psi)	205 ± 12	178 ± 11	136 ± 4	109 ± 3
% Elongation	133 ± 37	156 ± 39	168 ± 48	200 ± 58
Dk (barrers)	142.3	144.9	145.1	109.3
Example	5	6	7	8
Composition				
Prep 1	37.1	32.5	26.8	21.7
MBM	27.9	32.5	38.2	43.3
DMA	35	35	35	35
Darocur 1173	0.4	0.4	0.4	0.4
% of Diluent *	10	7	5	11
EWC(%)	26.1 ± 0.3	25.8 ± 0.3	25.8 ± 0.3	25.8 ± 0.1
Modulus (psi)	179 ± 5	215 ± 7	132 ± 6	101 ± 4
% Elongation	151 ± 42	106 ± 30	195 ± 65	179 ± 47
Dk (barrers)	118.8	129.6	116.5	107.9
Example	9	10	11	12
Composition				
Prep 1	35.4	31	25.5	20.7
MBM	26.6	31	36.5	41.3
DMA	38	38	38	38
Darocur 1173	0.4	0.4	0.4	0.4
% of Diluent *	12	7	7	5
EWC(%)	29.4 ± 0.3	30.0 ± 0.3	26.6 ± 0.2	26.7 ± 0.3
Modulus (psi)	215 ± 7	175 ± 7	132 ± 51	106 ± 4
% Elongation	99 ± 22	132 ± 40	166 ± 51	204 ± 55
Dk (barrers)	106.6	115.7	104.9	100.3
Example	13	14	15	16
Composition				
Prep 1	34.2	30	24.7	20
MBM	25.8	30	35.3	40
DMA	40	40	40	40
Darocur	0.4	0.4	0.4	0.4
% of Diluent *	12	11	8	9
EWC(%)	32.1 ± 0.1	31.2 ± 0.2	31.6 ± 0.3	31.7 ± 0.2
Modulus (psi)	218 ± 11	170 ± 6	131 ± 4	95 ± 3

% Elongation	110 \pm 34	130 \pm 51	185 \pm 53	203 \pm 47
Dk (barrers)	112.4	104.6	90.8	92.3

* % of the total reaction mixture of reactive components, and diluent.

5

Example 17

21.5% of α,ω -bismethacryloxypropyl polydimethylsiloxane with an average molecular weight of 5000 g/mol was combined with 42.5% MBM, 35% DMA and 1% DAROCUR 1173 and diluted with 3-methyl-3-pentanol to give a clear solution containing 22 weight % diluent. Lenses were made following the procedure of Example 1. The lens properties are shown in Table 2.

Example 18

15 Lenses were made using the procedure and reaction mixture described in Example 17, but with MPD in place of MBM. The lens properties are shown in Table 2.

Comparative Example 1

20 A reaction mixture was made using the formulation of Example 17, but with TRIS in place of MBM, and with 20% diluent. Lenses were made following the procedure of Example 1. The lens properties, shown in Table 2, show that the use of MBM (Example 17) or MPD (Example 18) gave lower moduli when used in place of TRIS.

25

Table 2 – Compositions and Properties of Silicone Hydrogel Polymers.

	Example 17	Comp. Ex. 1	Example 18
PDMS*	21.5	21.5	21.5
TRIS		42.5	
MBM	42.5		
MPD			42.5
DMA	35	35	35

Monomer/Diluent	78/22	80/20	78/22
Modulus	65±2 psi	87±3 psi	55±2 psi
Elongation at break	278±60%	307±88%	263±81%
Dk	110 barrers	147 barrers	75.6 barrers
EWC	28.2±0.3%	28.9±0.3%	31.0±0.3%

PDMS* = α,ω -bismethacryloxypropyl polydimethylsiloxane, ave. MW of 5000 g/mol

Example 19

29.0% of α,ω -bismethacryloxypropyl polydimethylsiloxane with an average molecular weight of 5000 g/mol was combined with 35% mono-methacryloxypropyl terminated PDMS (T1, Structure II, MW = 800 to 1000), 35% DMA and 1% DAROCUR 1173 and diluted with 3-methyl-3-pentanol to give a clear solution containing 23.0 weight % diluent. Lenses were made following the procedure of Example 1. The lens properties are shown in Table 3.

Example 20

29.0% of α,ω -bismethacryloxypropyl polydimethylsiloxane with an average molecular weight of 5000 g/mol was combined with 35% (3-methacryloxy-2-hydroxypropyloxy)propylbis(trimethylsiloxy)methylsilane (T2), 35% DMA and 1% DAROCUR 1173 and diluted with 3M3P to give a clear solution containing 37.6 weight % diluent. Lenses were made following the procedure of Example 1. The lens properties are shown in Table 3.

Table 3 - Compositions and Properties of Silicone Hydrogel Polymers.

	Example 19	Example 20
PDMS	29.0	29.0
T1	35.0	
T2		35.0
DMA	35.0	35.0
DAROCUR 1173	1.0	1.0
%/Diluent	23.0	37.6

Modulus	193±15 psi	175±11 psi
Elongation at break	87.9±42%	108±54%
Dk	171 barrers	94 barrers
EWC	31.1±0.2%	33.4±0.2%

The Examples show that the contact lenses made using the silicone-containing monomers of Structure I provide contact lenses which are clear and have a lower Young's modulus than the contact lenses made according to the Comparative Examples. A low modulus is desirable to provide contact lenses which are comfortable when worn.

Example 21

The following compositions were prepared, and cured with UV light into flat sheets. These sheets were extracted with isopropanol to remove diluent and any unreacted monomer, then equilibrated in isotonic borate buffered saline.

	A*	B*	C*	D	E
PDMS 1000 MW	11.41g	0g	0g	0g	0g
PDMS 3000 MW	0g	11.38g	0g	0g	0g
TRIS	0g	0g	11.38g	0g	0g
mPDMS 1000 MW **	0g	0g	0g	11.38g	0g
mPDMS 5000 MW ***	0g	0g	0g	0g	11.38g
DMA	6.13g	6.15g	6.13g	6.13g	6.13g
EGDMA	0g	0g	0.35g	0.37g	0.35g
Darocur 1173	0.08g	0.08g	0.08g	0.08g	0.08g
Diluent†	7.50g	10.40g	7.50g	7.50g	15.42g

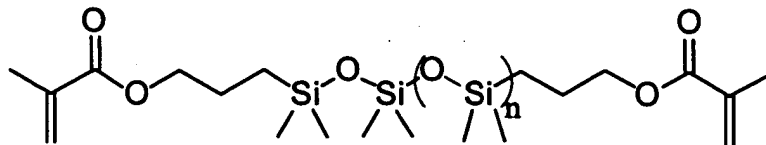
*For comparison, not according to the invention

** 800-1000MW was purchased from Gelest Inc. as "MCR-M11" brand mPDMS.

*** 5000MW was purchased from Gelest Inc. as "MCR-M17" brand mPDMS.

†3M3P

The structure of PDMS was:

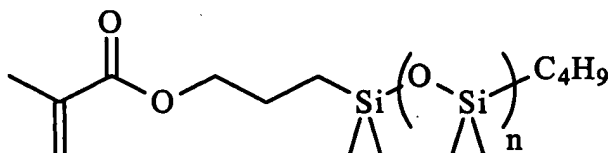


5

(Molecular weights for mPDMS and PDMS shown above are number average molecular weights).

10

The structure of mPDMS used in this example was:



15

For the determination of the properties after hydration, the 25 mm diameter hydrogel disks (each approximately 0.7 mm thick) were held between the 25 mm diameter parallel plates (plated with a crystal clad 80/100 grit coating) of a controlled stress rheometer (ATS Stresstech) with a vertical force of 10 N. The disks were immersed in water during the test to prevent dehydration. A stress sweep from 100 to 10,000 Pa at 1 Hz and 25°C was conducted on a disk of each material, to determine the range of the linear viscoelastic region for each formulation.

20

Once the limit of the linear viscoelastic region had been determined, the rheometer was set in frequency sweep mode using a stress less than the predetermined limit, and G' , G'' , and $\tan \delta$ of the 25 mm diameter hydrated disks were measured as a function of frequency from 0.01 – 30 Hz at several
5 temperatures (10, 25, 40 and 55°C), all the while maintaining a vertical force of 10 N on the hydrogel disks. The individual frequency scans of G' and $\tan \delta$ were then combined to form master curves for each material. The data for the shear modulus G' and $\tan \delta$ of the hydrogels at a reference temperature of 25°C are shown in tables 4 and 5.

10 The shear modulus of sample A was greater than B in the frequency range in which they were tested (as would be expected since the only difference between them is the molecular weight between crosslinks: 1000 vs. 3000), and their shear moduli gradually increased with increasing frequency. At the high frequency extreme, sample A appeared to be approaching a
15 transition, since the modulus appeared to be approaching a region of more rapid increase.

The $\tan \delta$ of samples A and B were below 0.2 for the most part, with the $\tan \delta$ of sample A increasing at the high frequencies in anticipation of a transition at higher frequencies.

20 Similarly, the shear modulus of sample D was greater than E in the frequency range in which they were tested (the only difference between them is the molecular weight of the dangling chains: 1000 vs. 5000), and their shear moduli gradually increased with increasing frequency.

The $\tan \delta$ of samples D and E was below 0.1 for the entire frequency
25 range in which they were tested, with the $\tan \delta$ of sample E below that of sample D.

The shear modulus of sample C increased rapidly in the frequency range in which it was tested, indicating a transition from a rubbery to a more rigid state. The bulky TRIS moiety reduced the internal molecular mobility of
30 the hydrogel (relative to the PDMS and mPDMS polymers), and caused the “glass” transition to shift to lower frequencies (\equiv higher temperatures).

The $\tan \delta$ of sample C reached a maximum at 100 Hz at the reference temperature of 40°C, indicative of the transition.

The addition of mPDMS was more effective than, the addition of di-capped PDMS in reducing the $\tan \delta$ of the hydrogel. The elastic properties of a material were controlled by the judicious addition of mPDMS while maintaining the homogeneity of the material. On the other hand, the addition of TRIS increased the $\tan \delta$ of a material.

Table 4

	G' (Pa)				
Frequency (Hz)	A	B	C	D	E
30.0000	5.84E+05	2.89E+05	2.03E+06	3.41E+05	2.22E+05
20.0000	5.70E+05	2.92E+05	1.81E+06	3.43E+05	2.24E+05
15.0000	5.56E+05	2.92E+05	1.63E+06	3.42E+05	2.23E+05
10.0000	5.37E+05	2.90E+05	1.42E+06	3.39E+05	2.24E+05
9.0000	5.34E+05	2.89E+05	1.37E+06	3.37E+05	2.24E+05
8.0000	5.27E+05	2.89E+05	1.35E+06	3.38E+05	2.24E+05
6.9971	5.22E+05	2.88E+05	1.29E+06	3.38E+05	2.22E+05
6.0000	5.16E+05	2.86E+05	1.22E+06	3.35E+05	2.22E+05
5.0000	5.08E+05	2.85E+05	1.13E+06	3.34E+05	2.22E+05
4.0000	4.96E+05	2.82E+05	1.07E+06	3.31E+05	2.20E+05
3.0000	4.84E+05	2.81E+05	9.88E+05	3.28E+05	2.20E+05
2.0000	4.74E+05	2.76E+05	8.49E+05	3.24E+05	2.20E+05
1.5000	4.63E+05	2.73E+05	7.85E+05	3.21E+05	2.20E+05
1.0000	4.48E+05	2.71E+05	6.96E+05	3.16E+05	2.19E+05
0.9000	4.46E+05	2.70E+05	6.72E+05	3.18E+05	2.19E+05
0.8000	4.43E+05	2.69E+05	6.62E+05	3.15E+05	2.17E+05
0.7001	4.39E+05	2.68E+05	6.33E+05	3.12E+05	2.18E+05
0.6000	4.35E+05	2.65E+05	6.07E+05	3.12E+05	2.18E+05
0.5000	4.32E+05	2.65E+05	5.78E+05	3.10E+05	2.17E+05
0.4000	4.25E+05	2.62E+05	5.48E+05	3.07E+05	2.16E+05
0.3000	4.18E+05	2.60E+05	5.06E+05	3.06E+05	2.15E+05
0.2000	4.07E+05	2.58E+05	4.59E+05	3.01E+05	2.14E+05
0.1500	4.00E+05	2.56E+05	4.31E+05	2.99E+05	2.18E+05
0.1000	3.91E+05	2.53E+05	3.93E+05	2.94E+05	2.12E+05
0.0900	3.90E+05	2.54E+05	3.82E+05	2.94E+05	2.12E+05
0.0800	3.86E+05	2.52E+05	3.75E+05	2.91E+05	2.13E+05
0.0700	3.86E+05	2.52E+05	3.63E+05	2.90E+05	2.13E+05
0.0600	3.83E+05	2.51E+05	3.52E+05	2.89E+05	2.12E+05
0.0500	3.80E+05	2.50E+05	3.39E+05	2.87E+05	2.12E+05
0.0400	3.75E+05	2.49E+05	3.27E+05	2.88E+05	2.13E+05
0.0300	3.74E+05	2.49E+05	3.07E+05	2.83E+05	2.11E+05
0.0200	3.64E+05	2.44E+05	2.87E+05	2.79E+05	2.09E+05
0.0150	3.60E+05	2.44E+05	2.76E+05	2.75E+05	2.07E+05
0.0100	3.53E+05	2.41E+05	2.58E+05	2.71E+05	2.08E+05

Table 5

	tan δ				
Frequency (Hz)	A	B	C	D	E
30.0000	0.2371	0.1084	0.5259	0.1025	0.0500
20.0000	0.2107	0.0964	0.5392	0.0869	0.0394
15.0000	0.1925	0.0917	0.5584	0.0767	0.0347
10.0000	0.1804	0.0917	0.5664	0.0725	0.0326
9.0000	0.1690	0.0877	0.5745	0.0671	0.0270
8.0000	0.1622	0.0871	0.5540	0.0704	0.0290
6.9971	0.1631	0.0876	0.5534	0.0702	0.0240
6.0000	0.1594	0.0883	0.5381	0.0682	0.0298
5.0000	0.1538	0.0872	0.5690	0.0656	0.0260
4.0000	0.1471	0.0839	0.5568	0.0635	0.0284
3.0000	0.1461	0.0879	0.5192	0.0633	0.0226
2.0000	0.1472	0.0902	0.5352	0.0607	0.0240
1.5000	0.1219	0.0892	0.5221	0.0654	0.0172
1.0000	0.1199	0.0953	0.5070	0.0648	0.0259
0.9000	0.1209	0.0915	0.4974	0.0553	0.0242
0.8000	0.1294	0.0978	0.4892	0.0601	0.0267
0.7001	0.1230	0.0898	0.4852	0.0647	0.0260
0.6000	0.1160	0.0953	0.4742	0.0640	0.0310
0.5000	0.1096	0.0939	0.4754	0.0664	0.0150
0.4000	0.1133	0.0969	0.4641	0.0626	0.0254
0.3000	0.1113	0.0976	0.4469	0.0651	0.0147
0.2000	0.1083	0.1019	0.4256	0.0653	0.0270
0.1500	0.1067	0.1047	0.4039	0.0699	0.0244
0.1000	0.1039	0.1066	0.3895	0.0744	0.0277
0.0900	0.1045	0.1084	0.3848	0.0761	0.0300
0.0800	0.1042	0.1066	0.3831	0.0774	0.0234
0.0700	0.0988	0.1076	0.3749	0.0709	0.0155
0.0600	0.0969	0.1112	0.3705	0.0755	0.0262
0.0500	0.1024	0.1137	0.3496	0.0765	0.0209
0.0400	0.0887	0.1117	0.3349	0.0782	0.0257
0.0300	0.0914	0.1160	0.3282	0.0830	0.0322
0.0200	0.1060	0.1185	0.3248	0.0792	0.0241
0.0150	0.1038	0.1270	0.2978	0.0753	0.0391
0.0100	0.1038	0.1326	0.2800	0.0846	0.0285

Example 22- SEALs Study

A double masked, contralateral, randomized, complete block clinical study was conducted to determine the relationship between SEALs and the Young's modulus of contact lenses. Lenses were made from two different silicone hydrogel compositions as follows.

Table 6

Component	Lens A (Wt %)	Lens B(Wt %)
Silicone based macromer	17.98	17.98
TRIS	21	14
mPDMS	21	28
DMA	25.5	26
blocked HEMA	5	5
PVP K90	5	5
TEGDMA	1.5	1

The silicone-based macromer refers to a prepolymer in which one mole was made from an average of 19.1 moles of 2-hydroxyethyl methacrylate, 2.8 moles of methyl methacrylate, 7.9 moles of methacryloxypropyltris(trimethylsiloxy)silane, and 3.3 moles of mono-methacryloxypropyl terminated mono-butyl terminated polydimethylsiloxane. The macromer was completed by reacting the aforementioned material with 2.0 moles per mole of 3-isopropenyl- ω,ω -dimethylbenzyl isocyanate using dibutyltin dilaurate as a catalyst.

Weight percentages are computed based on the total weight of all components; the balance of the compositions in Table 5 comprise initiators

and additives.

Lenses were made from these compositions having a nominal base curve of 8.5 mm and a diameter of 14.0 mm at 22°C. They had a nominal center thickness of 0.110 mm and a measured center thickness of 0.119 mm (Lens A) and 0.085 mm (Lens B). Both lenses were coated with a hydrophilic coating (PAA as in Example 37). Lens A had a Young's modulus of 109.4 psi and Lens B had a Young's modulus of 88.5 psi.

Subjects participating in the study were given a baseline examination and fitted with lenses made from the compositions shown in Table 6. They then wore the lenses for one week. Lenses were worn for daily wear. The subjects returned for a clinical evaluation of the presence of SEALs and other clinical data (e.g. visual acuity). Nineteen subjects (38 eyes) completed the study, eight of whom had a history of SEALs. Ten (10) eyes wearing Lens A exhibited SEALs. No eyes wearing Lens B exhibited SEALs.

Example 23 - SEALs Study

A study similar to that of Example 22 was conducted using lenses of different center thicknesses. The lenses had the following characteristics and gave the results shown in Table 7. In Table 7, column 4, "E" represents Young's modulus and "CT" represents the center thickness.

Table 7

Lens Type	Modulus (psi)	CT (μm)	E(CT) ² (psi • mm ²)	SEALs (%)
Etafilcon	40	110	0.48	1
Lens A*	110	124	1.69	10
Lens B*	88	105	0.97	0
Lens B*	88	170	2.54	24
Lotrafilcon A	238	80	1.52	5
Balafilcon A	155	90	1.26	5

*Refers to lens composition corresponding to Example 22.

This examples shows the combined effect of lens center thickness and modulus on SEALs.

Example 24 (Prophetic)

A set of lens characteristics is established for lenses that will not result in SEALs. This is accomplished by comparing the relative deflection $(k(Et^2)^{-1}$, where k = constant, E = Young's modulus, and t = center thickness of lenses having different moduli with those of the moduli and thicknesses of the lenses of Example 22 that did not induce SEALs. This range is used to establish mPDMS concentration ranges that will result in lenses whose use will not induce SEALs.

E (psi)	Thickness (μm)	[mPDMS] (wt %)*
45	<149	40
60	<129	30
100	<100	20
130	<88	5

*mPDMS concentration based on a composition comprised of the components of Lenses B in Example 22. One skilled in the art will recognize adjustments in mPDMS concentration when the lens is based on silicone hydrogels comprising additional or alternative components.

Example 25 (GTP Macromer Preparation)

Macromer A:

To a dry container housed in a dry box under nitrogen at ambient temperature was added 30.0 g (0.277 mol) of bis(dimethylamino)methylsilane, a solution of 13.75 ml of a 1M solution of TBACB (386.0 g TBACB in 1000 ml dry THF), 61.39 g (0.578 mol) of p-xylene, 154.28 g (1.541 mol) methyl methacrylate (1.4 equivalents relative to initiator), 1892.13 (9.352 mol) 2-(trimethylsiloxy)ethyl methacrylate (8.5 equivalents relative to initiator) and 4399.78 g (61.01 mol) of THF. To a dry, three-necked, round-bottomed flask equipped with a thermocouple and condenser, all connected to a nitrogen source, was charged the above mixture prepared in the dry box.

The reaction mixture was cooled to 15 °C while stirring and purging with nitrogen. After the solution reaches 15 °C, 191.75 g (1.100 mol) of 1-

trimethylsiloxy-1-methoxy-2-methylpropene (1 equivalent) was injected into the reaction vessel. The reaction was allowed to exotherm to approximately 62 °C and then 30 ml of a 0.40 M solution of 154.4 g TBACB in 11 ml of dry THF was metered in throughout the remainder of the reaction. After the temperature of
5 reaction reached 30 °C and the metering began, a solution of 467.56 g (2.311 mol) 2-(trimethylsiloxy)ethyl methacrylate (2.1 equivalents relative to the initiator), 3636.6 g (3.463 mol) n-butyl monomethacryloxypropyl-polydimethylsiloxane (3.2 equivalents relative to the initiator), 3673.84 g (8.689 mol) TRIS (7.9 equivalents relative to the initiator) and 20.0 g
10 bis(dimethylamino)methylsilane was added.

The mixture was allowed to exotherm to approximately 38-42 °C and then allowed to cool to 30 °C. At that time, a solution of 10.0 g (0.076 mol) bis(dimethylamino)methylsilane, 154.26 g (1.541 mol) methyl methacrylate (1.4
15 equivalents relative to the initiator) and 1892.13 g (9.352 mol) 2-trimethylsiloxy)ethyl methacrylate (8.5 equivalents relative to the initiator) was added and the mixture again allowed to exotherm to approximately 40 °C. The reaction temperature dropped to approximately 30 °C and 2 gallons of THF were added to decrease the viscosity. A solution of 439.69 g water, 740.6 g
20 methanol and 8.8 g (0.068 mol) dichloroacetic acid was added and the mixture refluxed for 4.5 hours to de-block the protecting groups on the HEMA. Volatiles were then removed and toluene added to aid in removal of the water until a vapor temperature of 110 °C was reached.

25 The reaction flask was maintained at approximately 110 °C and a solution of 443 g (2.201 mol) TMI and 5.7 g (0.010 mol) dibutyltin dilaurate were added. The mixture was reacted until the isocyanate peak was gone by IR. The toluene was evaporated under reduced pressure to yield an off-white, anhydrous, waxy reactive monomer. The macromer was placed into acetone
30 at a weight basis of approximately 2:1 acetone to macromer. After 24 hrs,

water was added to precipitate out the macromer and the macromer was filtered and dried using a vacuum oven between 45 and 60 °C for 20-30 hrs.

Macromer B:

- 5 The procedure for Macromer A used except that 19.1 mole parts HEMA, 5.0 mole parts MAA, 2.8 mole parts MMA; 7.9 mole parts TRIS, 3.3, mole parts mPDMS, and 2.0 mole parts TMI were used.

Macromer C:

- 10 The procedure for Macromer A was used except that 19.1 mole parts HEMA, 7.9 mole parts TRIS, 3.3 mole parts mPDMS, and 2.0 mole parts TMI were used.

Examples 26 - 36 (Lens Formation)

- 15 Hydrogel were made from the monomer mixtures shown on Table 8. All amounts are calculated as weight percent of the total weight of the combination with the balance of the mixture being minor amounts of additives.

Polymerization was conducted in the presence of the diluents listed.

- 20 Contact lenses were formed by adding about 0.10 g of the monomer mix to the cavity of an eight cavity lens mold of the type described in U.S. Patent 4,640,489 and curing for 1200 sec. Polymerization occurred under a nitrogen purge and was photoinitiated with visible light generated with a Philips TL 20W/03T fluorescent bulb. After curing, the molds were opened, and the lenses were either released in a 1:1 blend of water and ethanol, then leached
25 in ethanol to remove any residual monomers and diluent, or released in a 60% IPA/water, then leached in IPA/DI to remove any residual monomers and diluent. Finally the lenses were equilibrated in physiological borate-buffered saline. The lenses had the properties described in Table 8

Table 8

EXAMPLE	26	27	28	29	30	31	32	33	34	35	36
Macromer	B	A	A	C	C	A	A	A	A	A	A
Macromer	30.00	17.98	25.00	60.00	20.00	17.98	17.98	19.98	17.98	17.98	19.98
TRIS	0.00	14.00	18.00	0.00	40.00	21.00	21.00	8.00	20.00	25.00	20.00
DMA	27.00	26.00	28.00	36.00	36.00	25.50	25.50	26.00	22.00	9.00	23.00
MPDMS	39.00	28.00	18.00	0.00	0.00	21.00	21.00	28.50	25.50	30.00	28.50
Norbioc	2.00	2.00	2.00	3.00	3.00	2.00	2.00	2.00	2.00	2.00	2.00
CGI 1850	2.00	1.00	1.00	1.00	1.00	1.00	1.00	1.00	1.00	1.00	1.00
TEGDMA	0.00	1.00	0.00	0.00	0.00	1.50	1.50	1.50	1.50	0.50	1.50
HEMA	0.00	5.00	0.00	0.00	0.00	5.00	5.00	5.00	5.00	7.00	5.00
Blue	0.00	0.02	0.00	0.00	0.00	0.02	0.02	0.02	0.02	0.02	0.02
HEMA											
PVP (K90)	0.00	5.00	8.00	0.00	0.00	5.00	5.00	8.00	5.00	7.50	9.00
Diluent %	41	20	20	20	None	20	50.00	37.50	20.00	40.00	50.00
Diluent	3M3P	*D3O	3M3P	3M3P	NA	D3O	TAA	3M3P	ethyl lactate	3M3P	3M3P
%EWC	49.2	39.1	48.5	40.9	37.1						
Modulus (psi)	73	85.3	59	273	102						
%	200	251	261	74	384						
Elongation											
Dk (edge corrected)	109.4	109	97.9	34.5	79.8						

*With 5% HOAc added

Example 37

To lenses from Example 27 immersed in a solution of 1.0% 250,000 M_w polyacrylic acid in water at 45°C was added 0.1% 1-[3-(dimethylamino)propyl]-3-ethylcarbodiimide hydrochloride. After stirring for 30 minutes the lenses are rinsed in borate-buffered saline solution. The dynamic contact angles of the resulting poly(sodium acrylate)-coated lenses are 44° advancing and 42° receding.

Comparative Example 2

Lenses were made by curing a blend of 57.5% TRIS, 40.0 % DMA, 1.5% 1,3-bis(3-methacryloxypropyl)tetrakis(trimethylsiloxy)disiloxane and 1.0% 2-hydroxy-2-methyl-1-phenyl-propan-1-one (by weight) in contact lens molds under UV light. The lenses were released into ethanol and transferred to borate-buffered saline solution. The lenses had properties given in Table 9, but when extended they returned to their original shape very slowly due to their high tan δ . In fact, even during ordinary lens handling, these lenses typically did not retain a their symmetrical shape. Further, they had less than desirable Dks and were very unwettable.

Table 9 – Properties of Lenses From Comparative Example 2.

Tensile Modulus (psi)	67
Elongation at break (%)	674
Water Content (%)	40.2
Edge corrected Dk (barrers)	70.2